

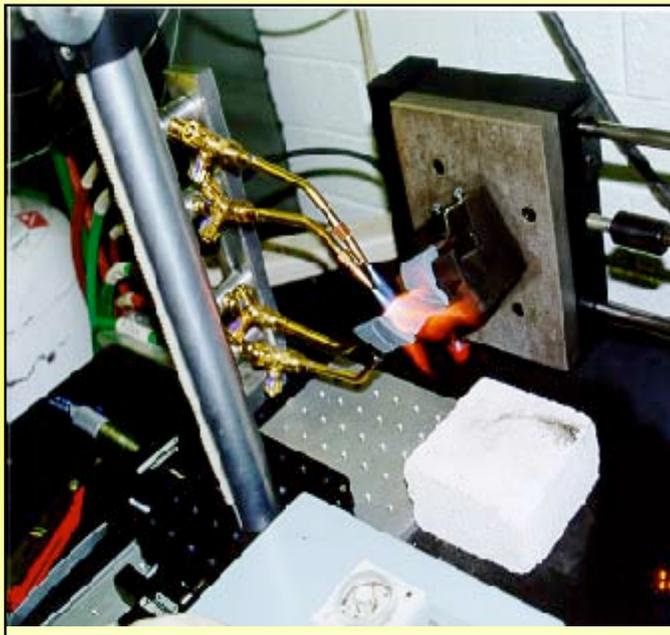
Effect of Material Parameters on Mechanical Performance of Environmental Barrier Coatings

*H. T. Lin, A. A. Wereszczak, M. K. Ferber, and
T. P. Kirkland*

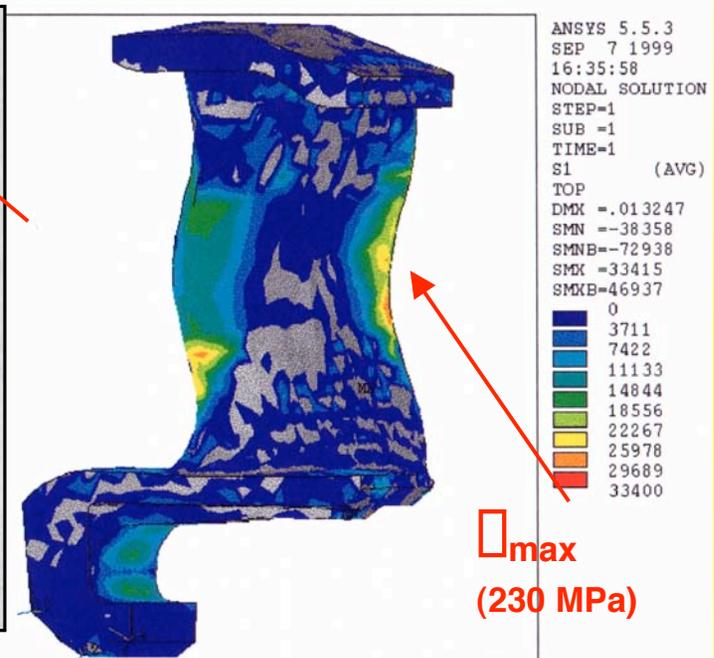
Ceramic Science and Technology Group
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Oak Ridge National Laboratory
Oak Ridge, TN 37831-6068

Research sponsored by U.S. DOE, Microturbine Materials Program, under Contract DE-AC05-00OR22725 with UT-Battelle, LLC.

Study was Motivated by Fracture of EBC-SN282 Nozzle During Thermal Shock Proof Test at Solar Turbines



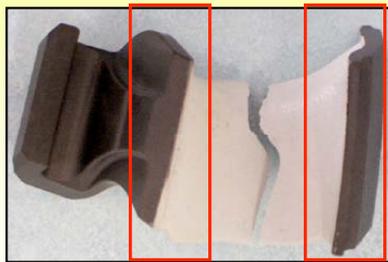
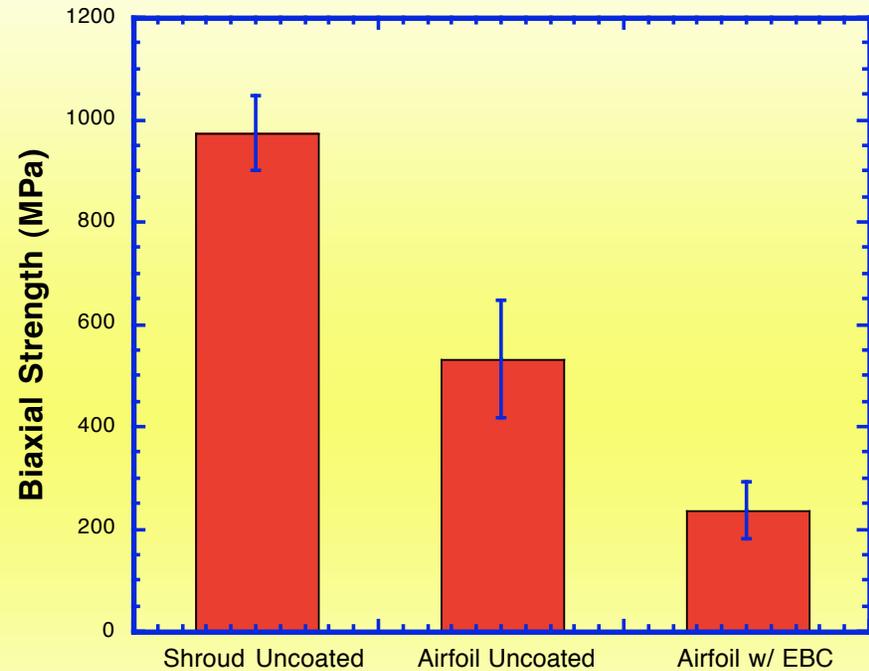
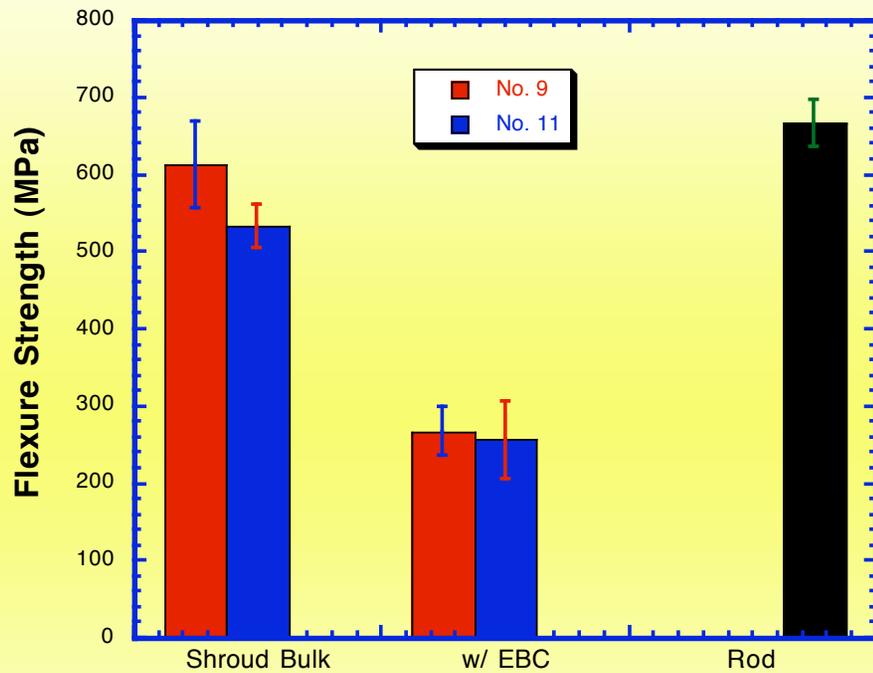
Solar SN282_11
2072°F, 28.6 s



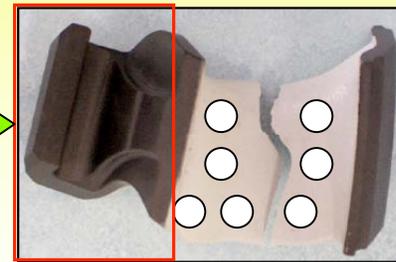
Suction side

The first stage nozzle is designed for Centaur 50S Gas Turbine

Significant Strength Degradation Observed in EBC SN282 Samples Extracted from Nozzles

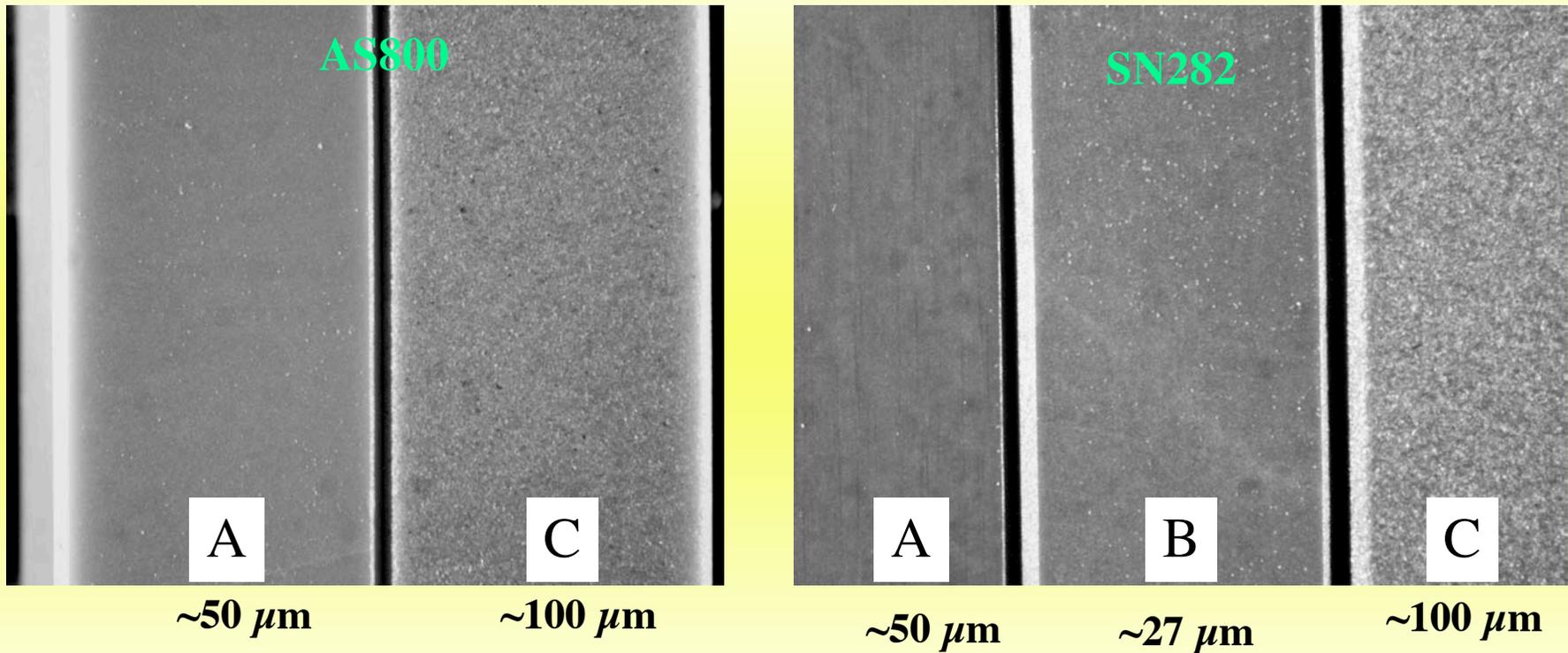


Inner
Shroud



The strength of bend bars from inner shroud region is comparable to those obtained from production rods

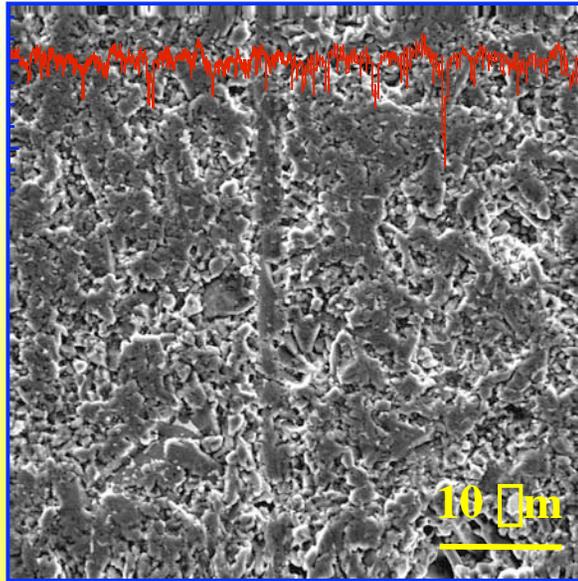
Grit Blast Conditions Employed for Surface Preparation Result in Different Surface Morphology of Substrate



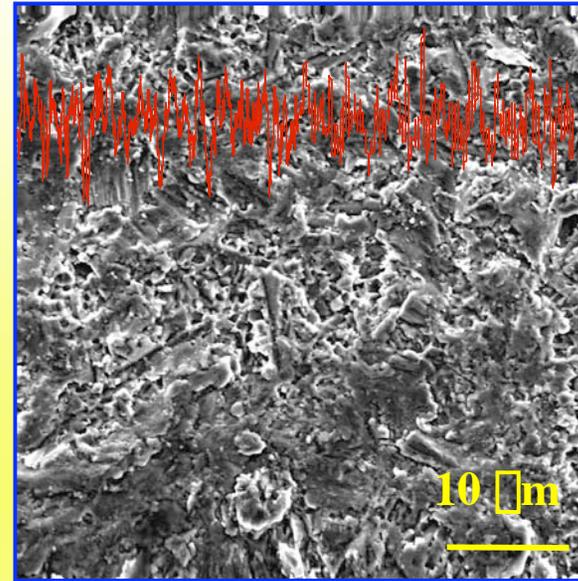
Grit blast step was carried out prior to the EBC coating process to ensure a better mechanical adhesion

Grit Blast Conditions Employed Could Introduce Substantial Surface (and Subsurface) Damages

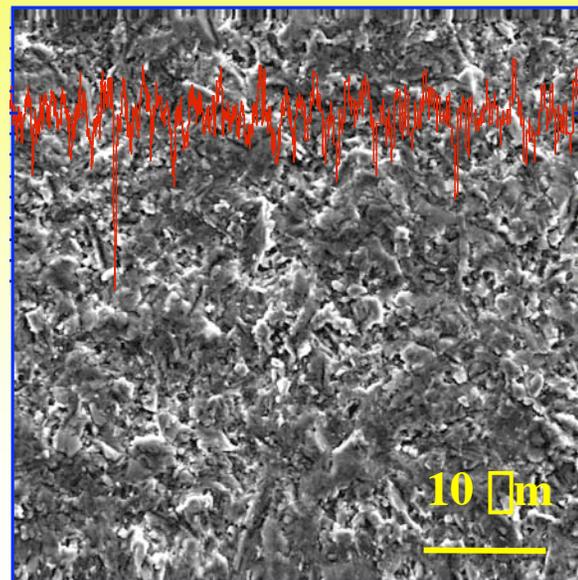
As-machined



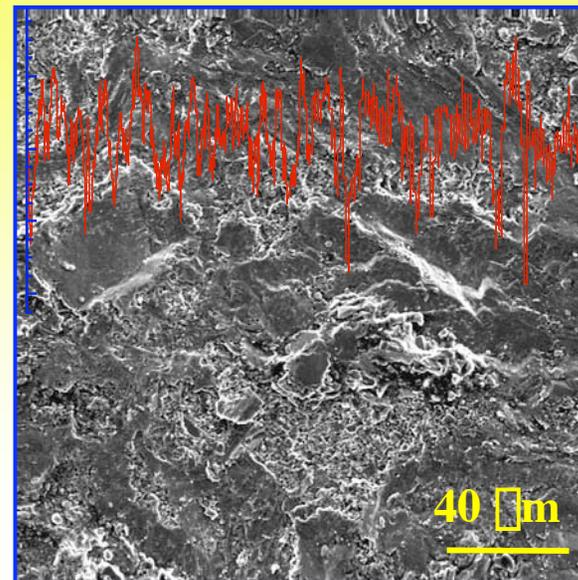
Condition "A" (~ 50 μm)



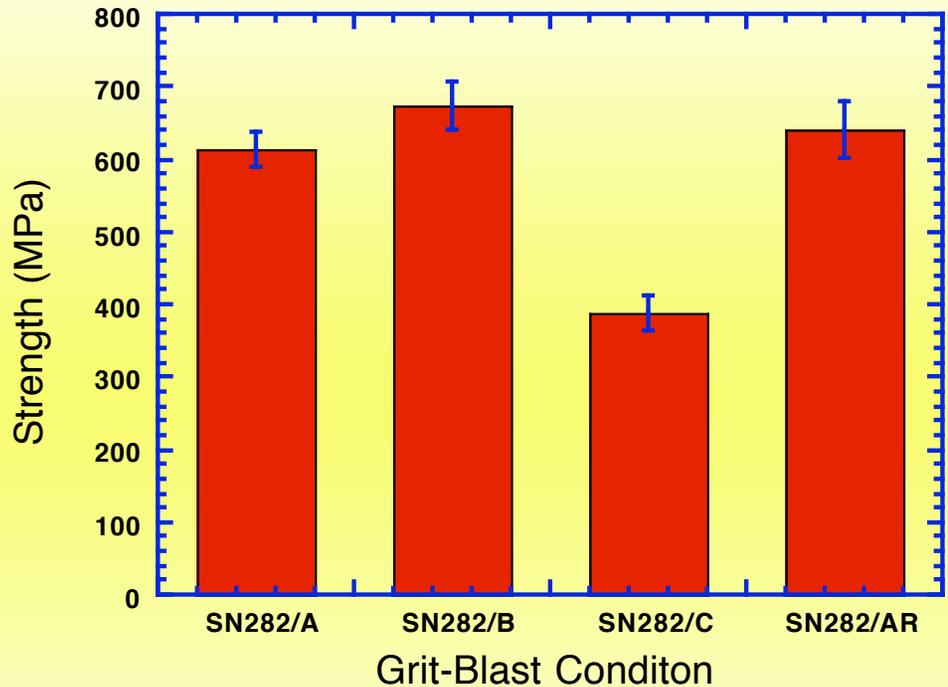
Condition "B" (~ 27 μm)



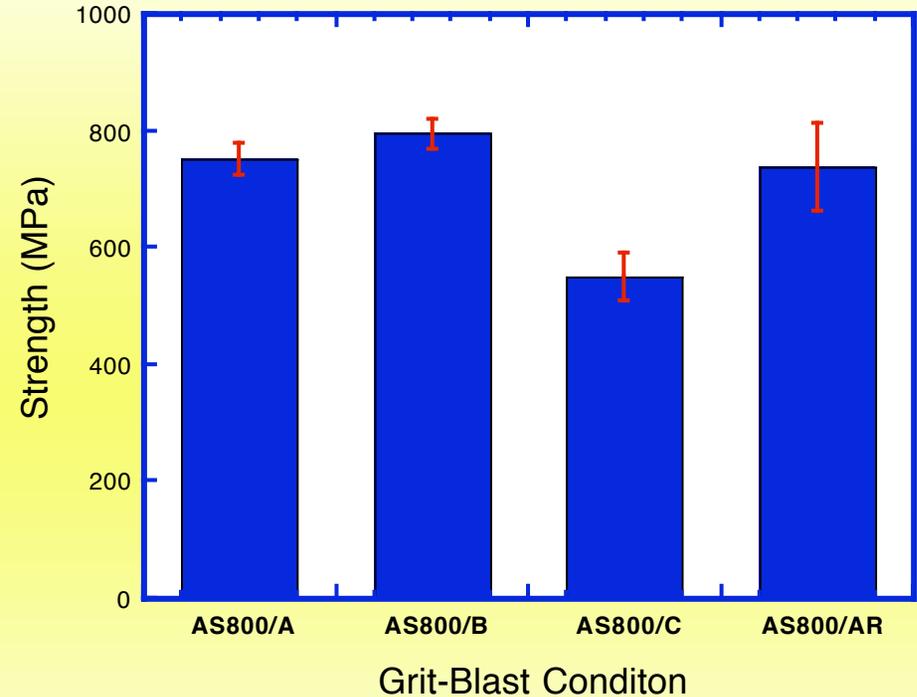
Condition "C" (~ 100 μm)



Grit Blast Conditions Employed for Surface Preparation Could Influence the Mechanical Property of Airfoils



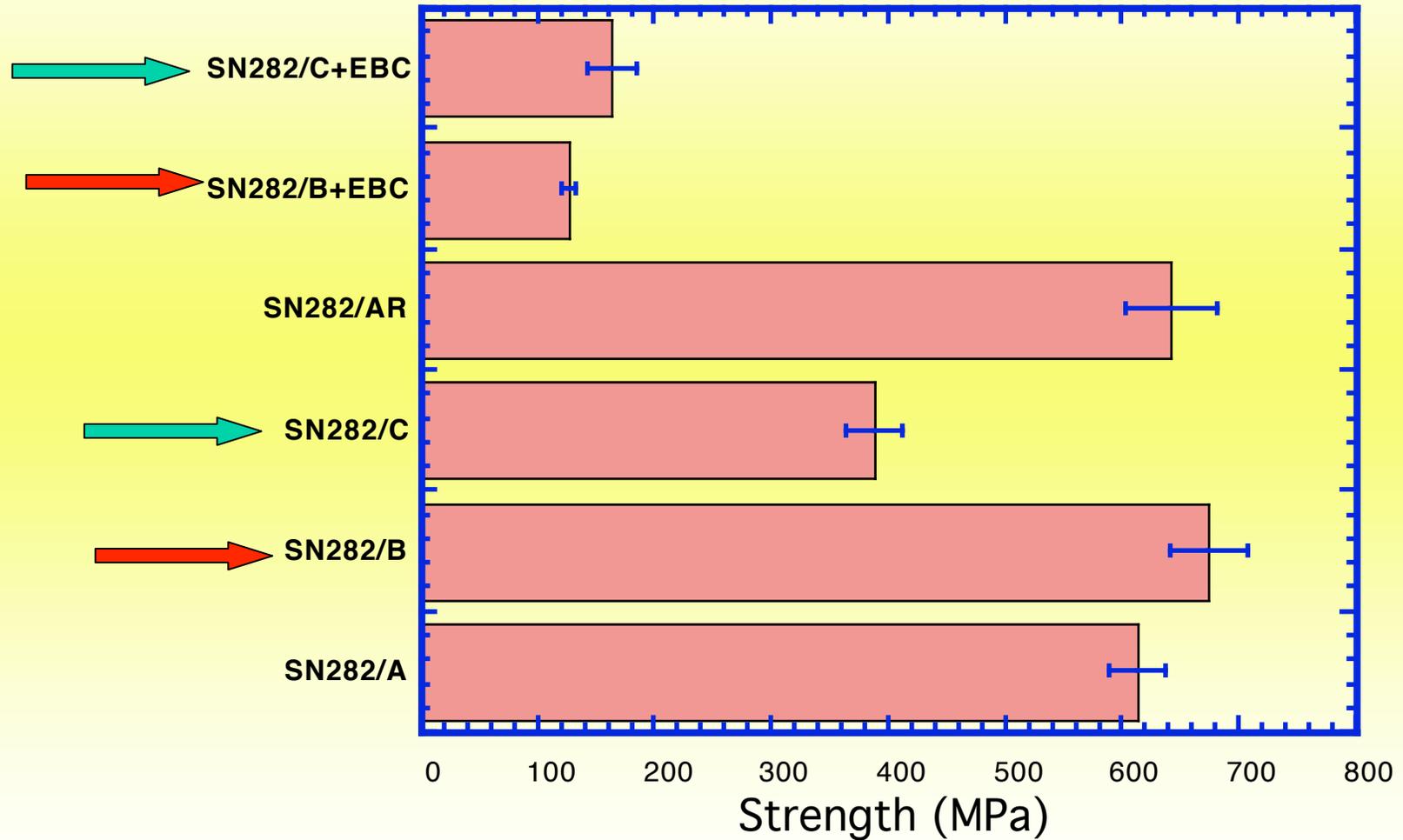
K_{IC} (SN282):
6 MPa \sqrt{m} \square 40% decrease



K_{IC} (AS800):
8.5 MPa \sqrt{m} \square 26% decrease

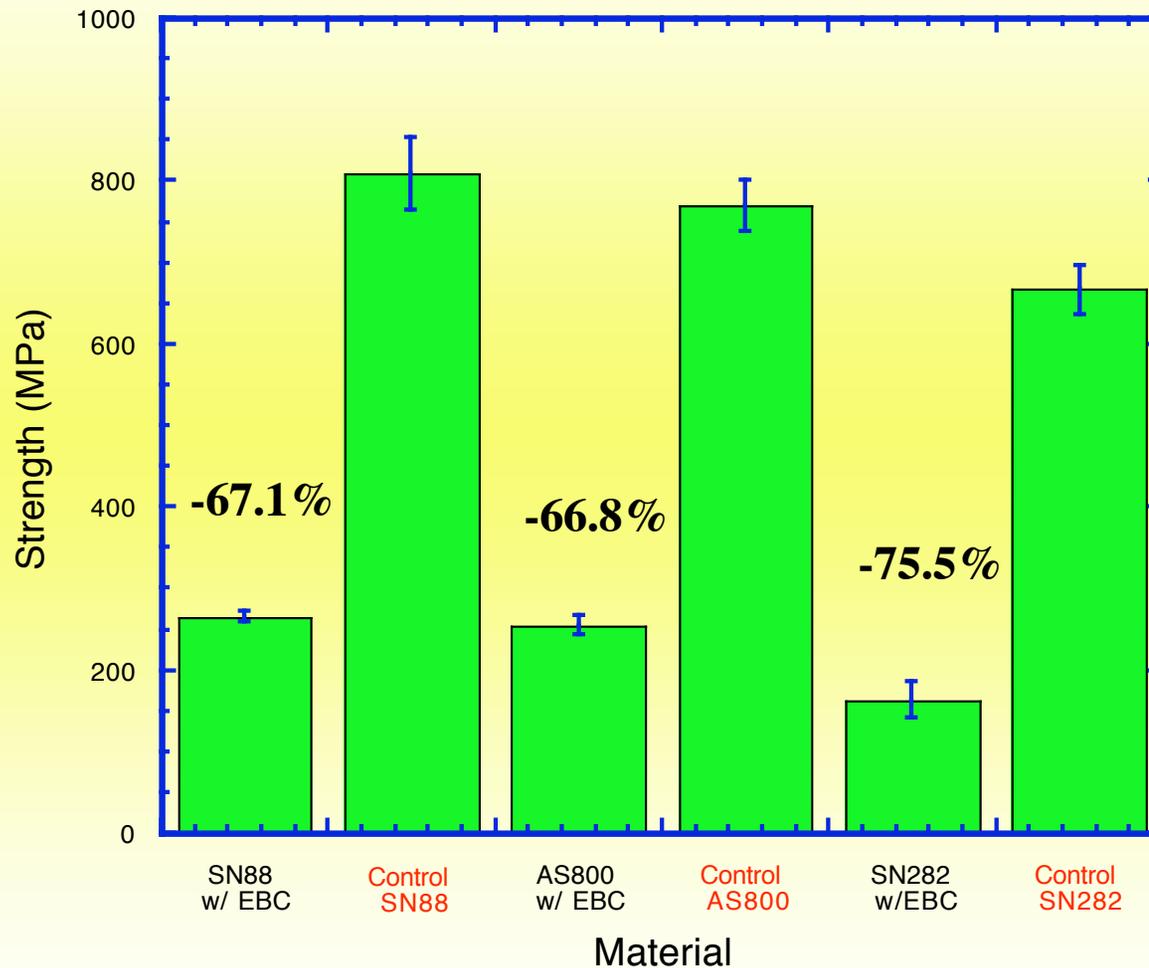
**The extent of strength degradation is related to the material toughness
(damage tolerance) and also hardness**

Mechanical Strength of SN282 Samples is Significantly Decreased After Deposition of BSAS EBC System



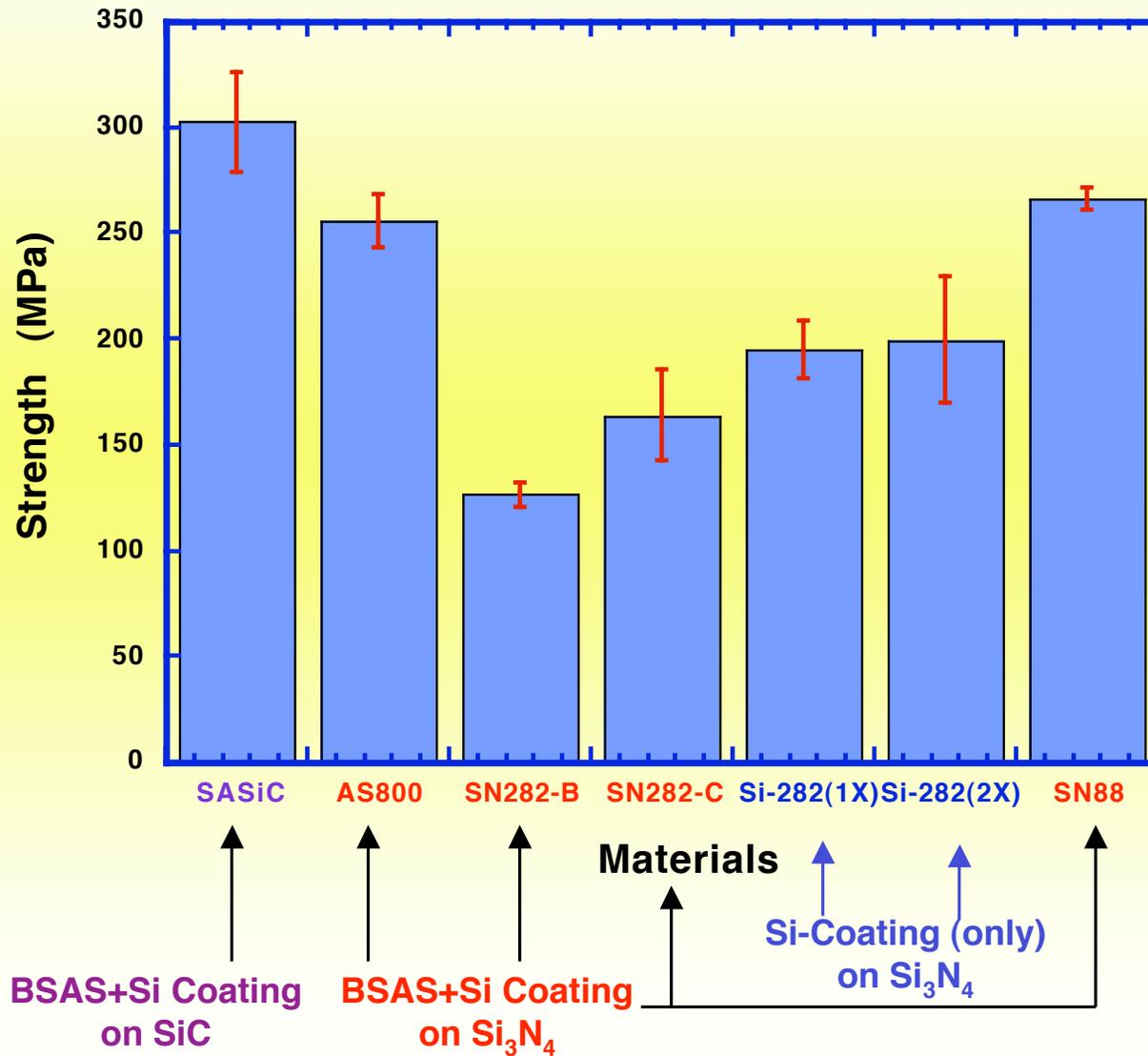
Strength degradation of EBC Si_3N_4 is independent upon grit blast condition

Substantial Strength Degradation Occurred in all EBC-Coated Commercially Available Advanced Silicon Nitrides



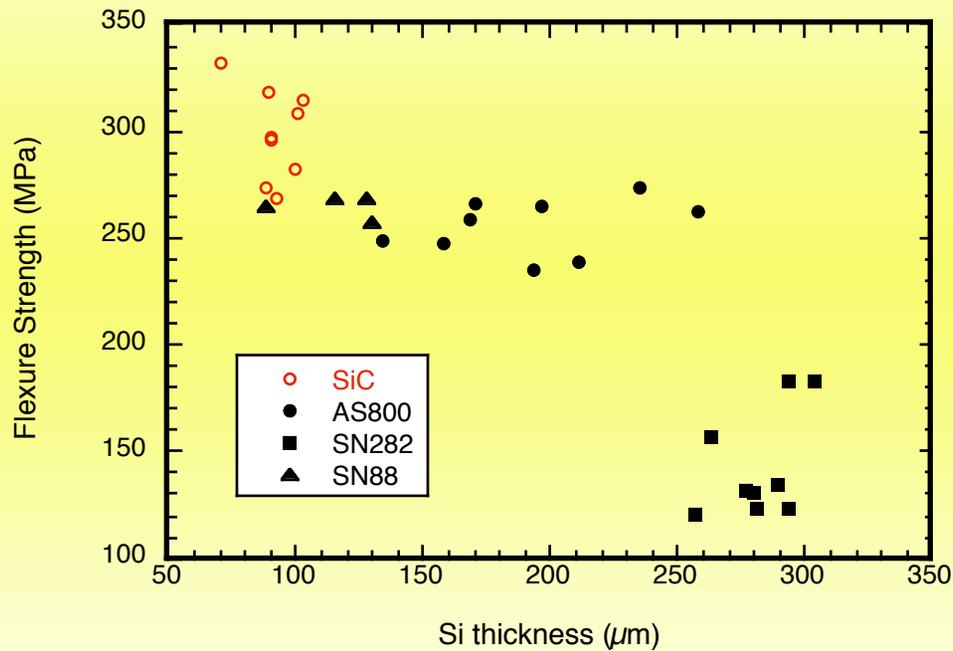
Extent of strength degradation is independent of mechanical properties (strength & toughness) of substrates

Flexural Strength Comparison of BSAS EBC Coated Silicon Nitride Ceramic Samples

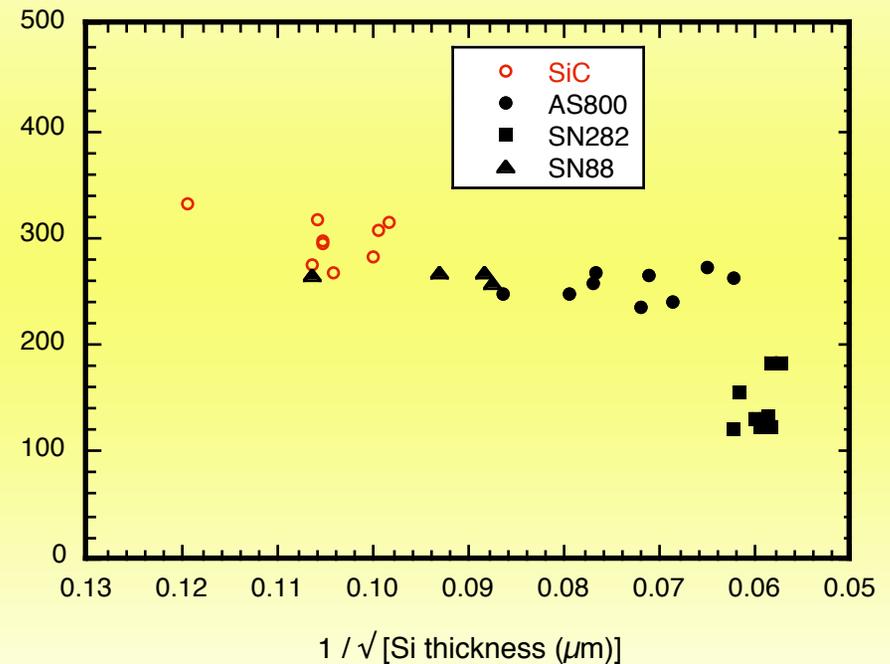


Dependence of BSAS EBC/Si₃N₄ Strength on Si Bond Coat Thickness Appears to be Very Subtle

Strength as a Function of Si Thickness



Strength as a Function of 1 / √ (Si Thickness)



$$K_{IC} = \sigma_a Y \sqrt{a}$$

There Was no Conclusive Dependence of Si/Si₃N₄ Strength on Si Thickness

Specimen ID	Max Stress (MPa)	Si coating thickness (μm)	Ra (μm)	Rq (μm)	Rp (μm)	Rv (μm)	Rt (μm)	Rz (μm)
1X-4	172.9	168	7.51	9.47	22.18	20.97	55.78	43.15
1X-3	193.0	93	7.76	9.92	26.97	20.14	61.15	47.11
1X-5	197.3	113	7.82	9.97	24.39	22.40	64.26	46.79
1X-1	201.6	133	9.49	11.81	27.51	24.42	64.99	51.93
1X-2	209.9	145	8.05	9.57	19.65	19.32	45.86	38.96
Ave	194.9	130.4	8.13	10.15	24.14	21.45	58.41	45.59
Std Dev	13.8	28.9	0.79	0.95	3.30	2.01	7.90	4.84
2X-6	175.5	264	7.29	9.19	19.52	20.26	60.00	39.78
2X-8	177.1	221	7.27	8.95	20.66	18.73	45.99	39.39
2X-7	189.9	290	8.52	11.01	33.58	21.69	73.05	55.27
2X-9	208.8	298						
2X-10	247.0	269	7.87	9.60	23.39	19.22	53.81	42.61
Ave	199.7	268.4	7.74	9.69	24.29	19.98	58.21	44.26
Std Dev	29.6	30.0	0.59	0.92	6.40	1.31	11.43	7.48



Flexure strength was essentially the same even though coating thickness was doubled

There May Appear to be Some Dependence of Si/Si₃N₄ Strength on Si Roughness Thickness (Ra & Rq)

Specimen ID	Max Stress (MPa)	Si coating thickness (µm)	Ra (µm)	Rq (µm)	Rp (µm)	Rv (µm)	Rt (µm)	Rz (µm)
1X-4	172.9	168	7.51	9.47	22.18	20.97	55.78	43.15
1X-3	193.0	93	7.76	9.92	26.97	20.14	61.15	47.11
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1X-1	201.6	133	9.49	11.81	27.51	24.42	64.99	51.93
1X-2	209.9	145	8.05	9.57	19.65	19.32	45.86	38.96
Ave	194.9	130.4	8.13	10.15	24.14	21.45	58.41	45.59
Std Dev	13.8	28.9	0.79	0.95	3.30	2.01	7.90	4.84
2X-6	175.5	264	7.29	9.19	19.52	20.26	60.00	39.78
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Std Dev	29.6	30.0	0.59	0.92	6.40	1.31	11.43	7.48

When the roughness parameters shown above are used to “back-calculate” the crack geometry correction factor, Y, there is good correlation with Ra and Rq and surface-located, elliptical, strength-limiting flaws shown below

	1x	2x
Ra	1.80	1.80
Rq	1.61	1.61
Rp	1.05	1.02
Rv	1.11	1.12
Rt	0.67	0.66
Rz	0.76	0.75

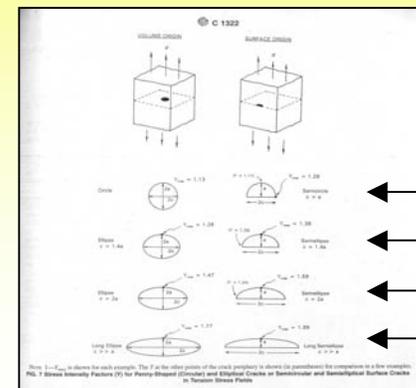
Good Correlation

No physical meaning

$$S = \frac{K_{Ic}}{Y\sqrt{c}}$$

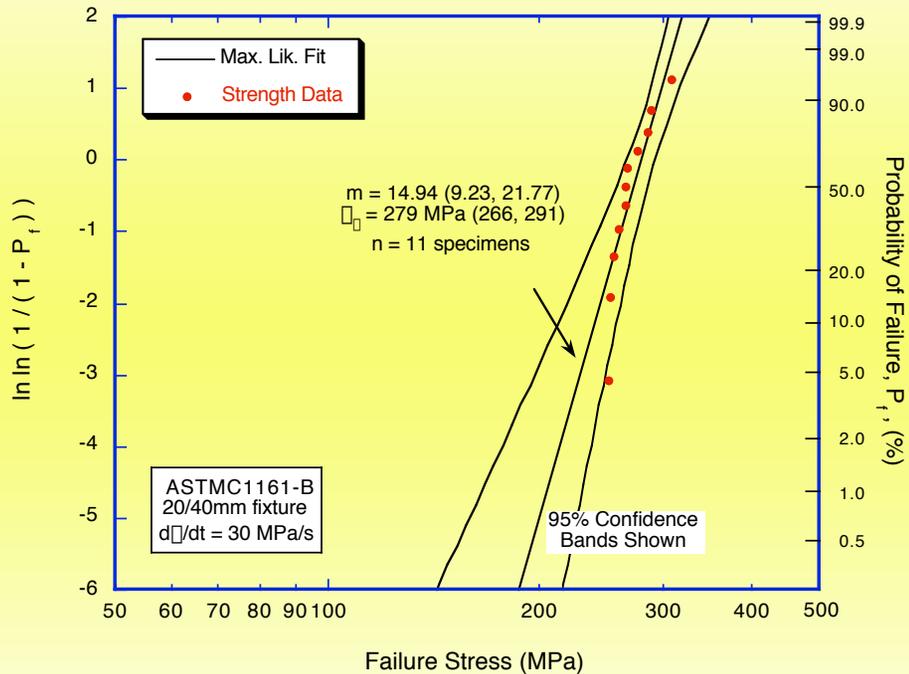
or

$$Y = \frac{K_{Ic}}{S\sqrt{c}}$$



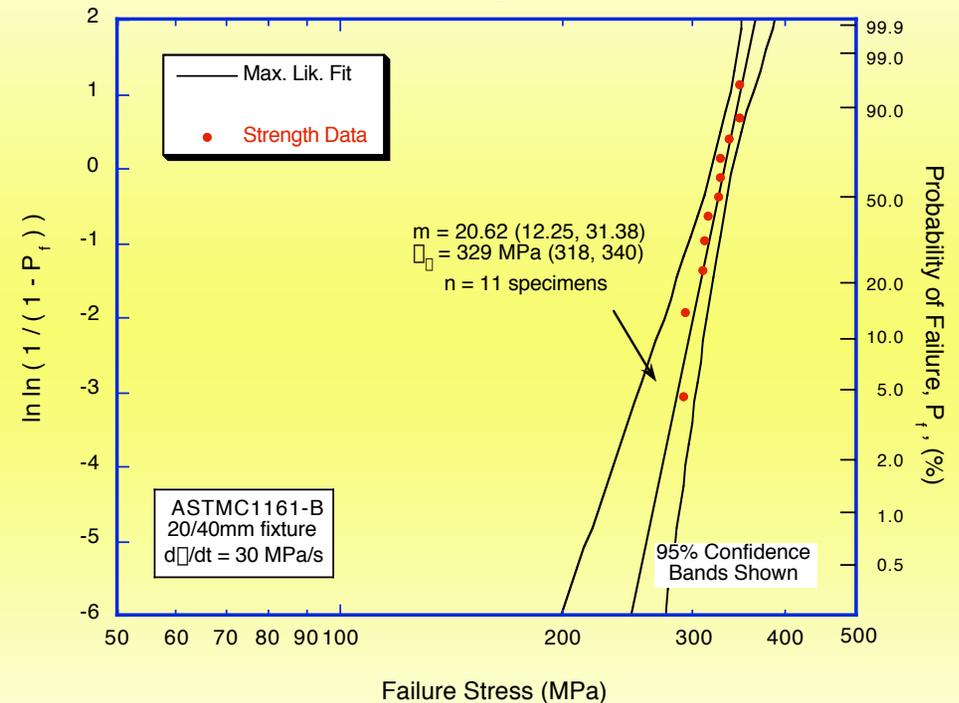
However, Si-SN282 Strength Results Seems to Exclude the Hypothesis of “Si” Surface Roughness Effect

Si-SN282-As-Coated
Uncensored Flexure Strength Distribution
20°C - 30 MPa/s



$$\sigma_{\text{avg}} = 253 \pm 9.8 \text{ MPa}$$

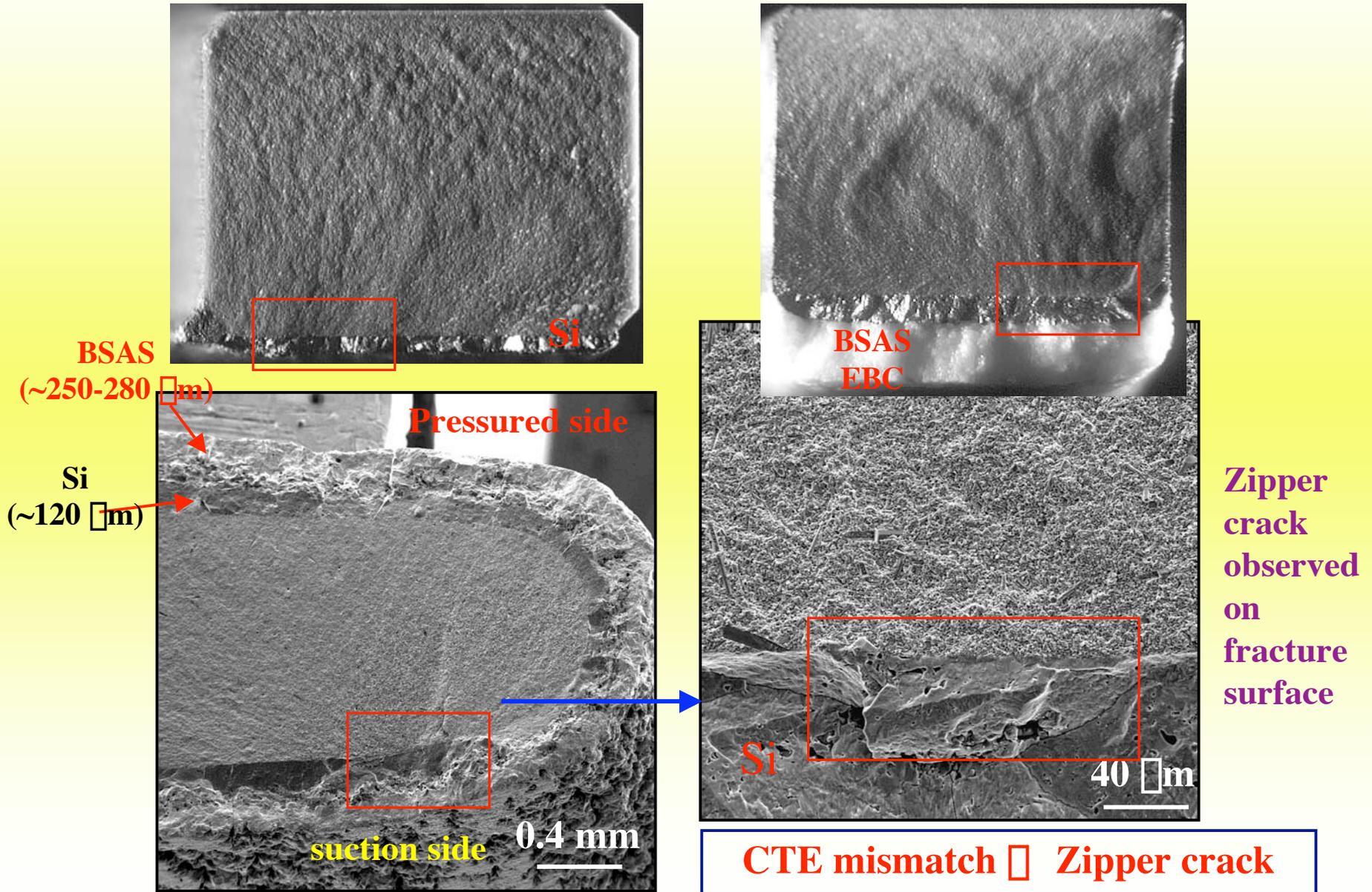
Si-SN282-600 Grit
Uncensored Flexure Strength Distribution
20°C - 30 MPa/s, Logitudinally Machined



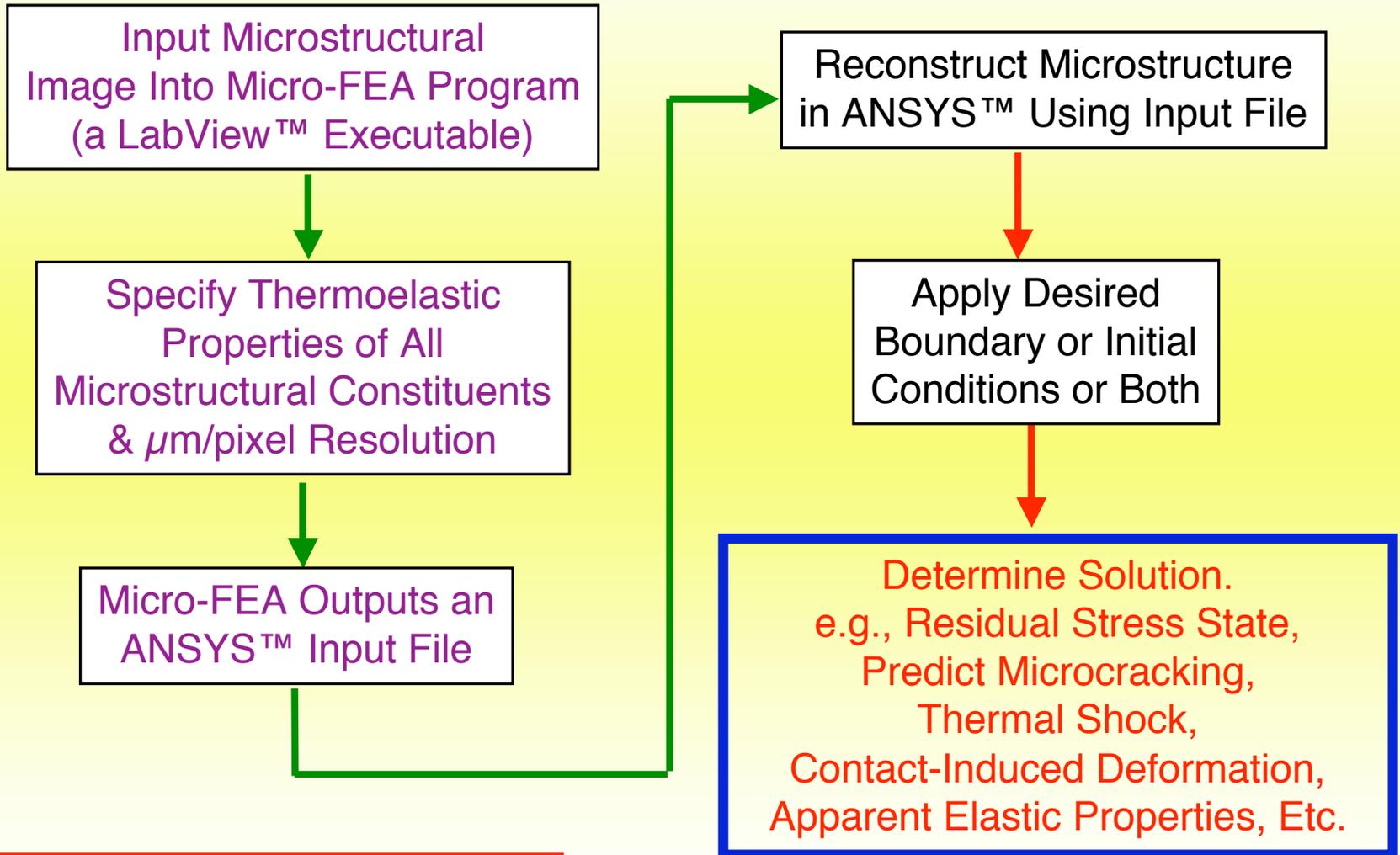
$$\sigma_{\text{avg}} = 270 \pm 17.7 \text{ MPa}$$

Fine surface finish (600 grit) of Si bond coat slightly improve strength and Weibull modulus

Zipper Crack Observed in all Si- and BSAS EBC-Coated Silicon Nitride Samples



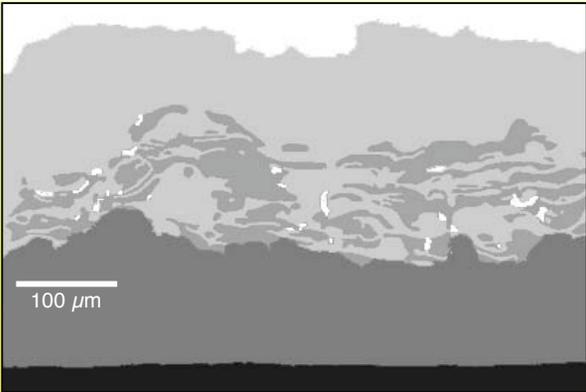
Micro-FEA (Finite Element Analysis) Method



By Wereszczak & Ferber

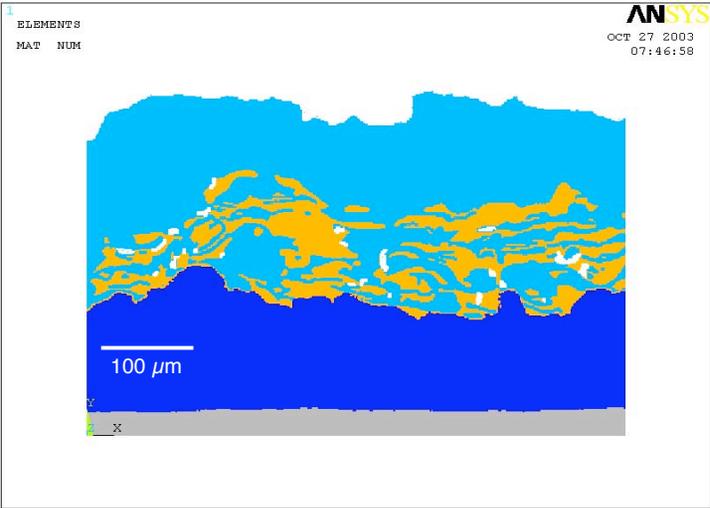
Micro-FEA of SN282 Bend Bar Coated with BSAS EBC

Microstructural Region



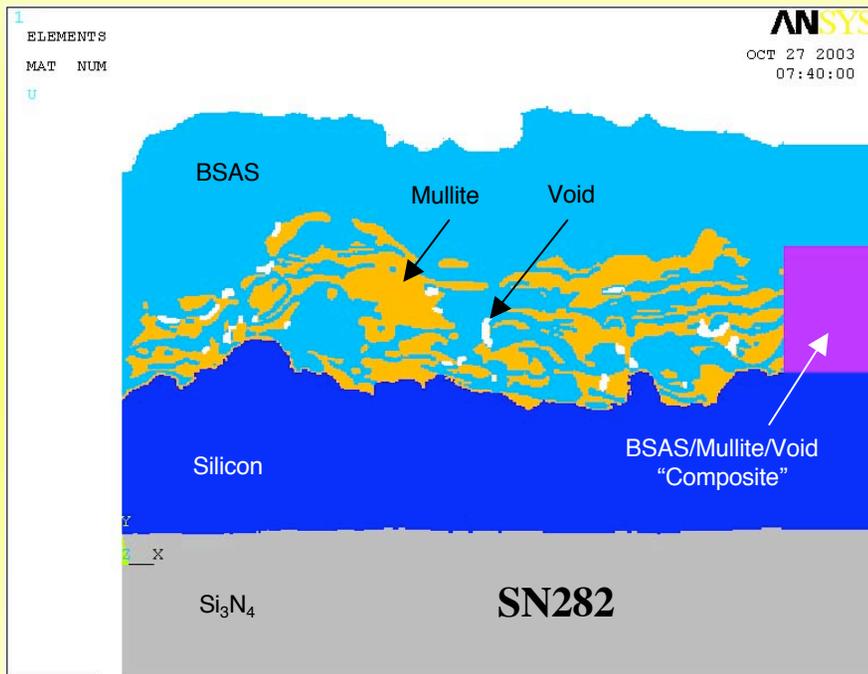
281 x 400 pixels
1.45 μm/pixel
407.45 x 580.00 μm

μ-FEA Reconstructed Model



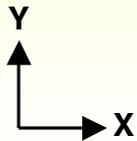
Thermoelastic Properties of BSAS EBC System and SN282 Silicon Nitride Substrate

μ -FEA Reconstructed Model



	E (GPa)	ν	CTE (ppm/C)
Si ₃ N ₄	315	0.29	2.9
Silicon	80	0.20	4.9
BSAS	80	0.24	5.0
Mullite	130	0.23	3.5
BSAS/Mullite	60	0.23	4.2
TREF = 1250C			
TUNIF = 25C			

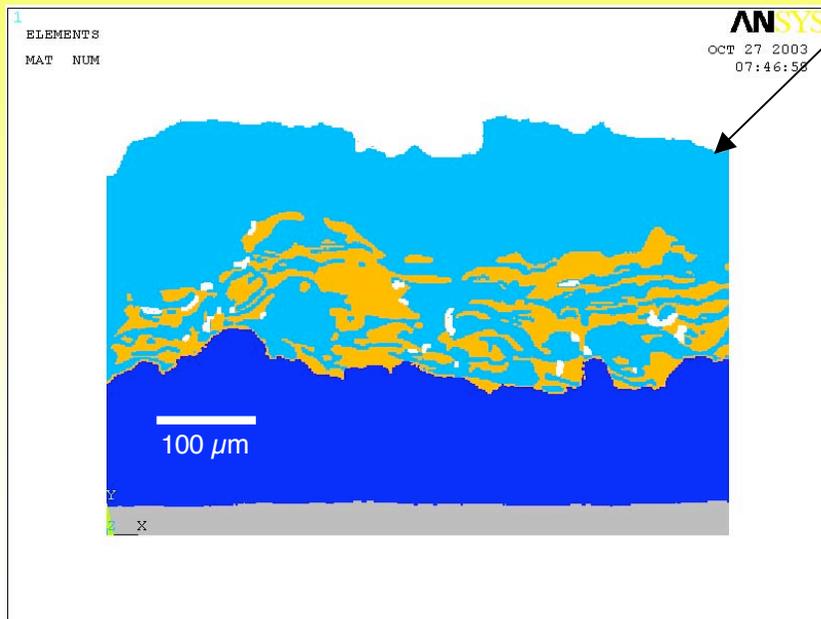
BSAS EBC properties by
Tania Bhatia @ UTRC



Inclusion of a Semi-Infinite Field in the FEA Acts To Apply Appropriate BCs on the Micro-Modeled Region

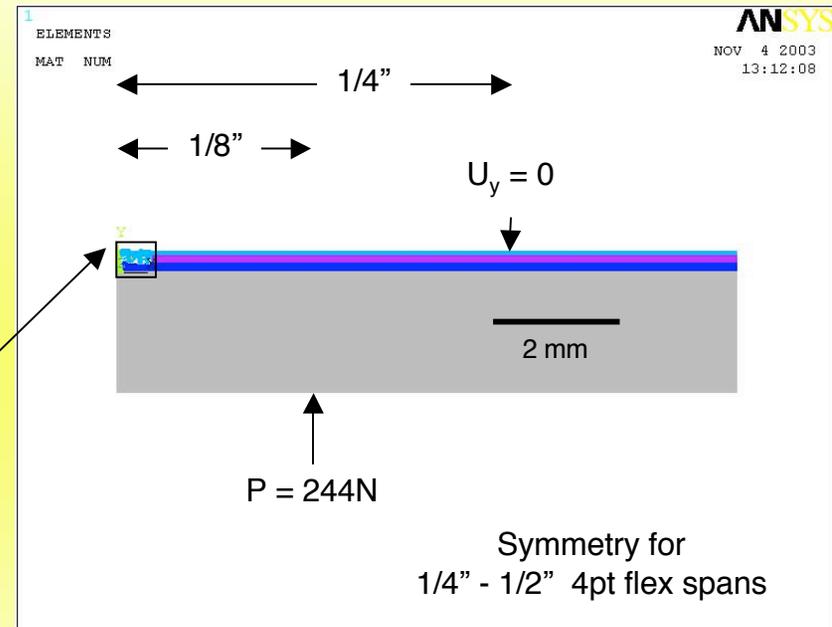
SEM micrograph taken from a polished cross section of bend bar extracted from EBC-SN282 nozzle

μ -FEA Reconstructed Model

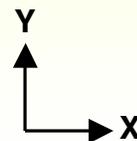


Detail

Whole Modeled Geometry

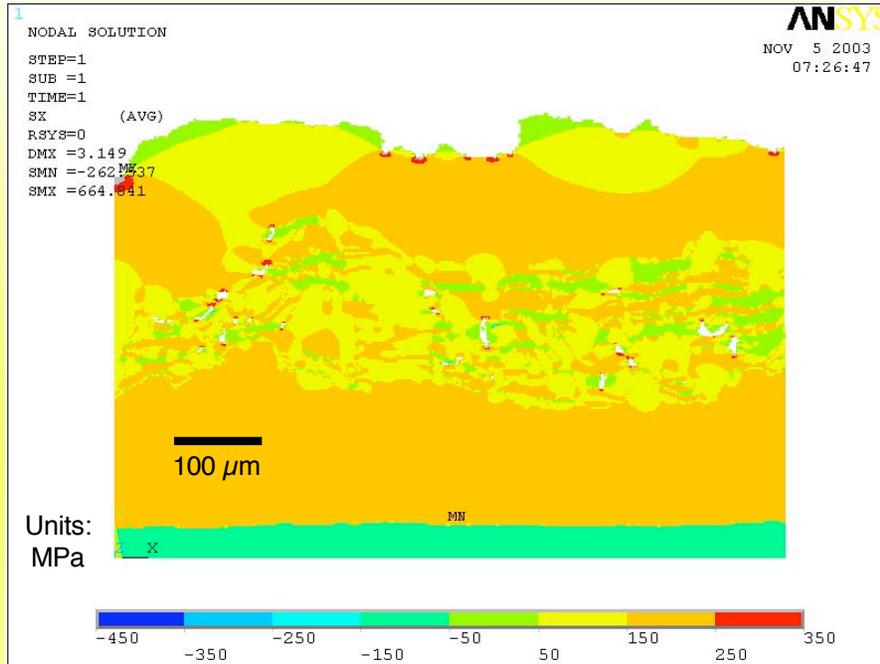


Case study was also carried out on ASTM standard MOR bar (20/40 mm loading spans)

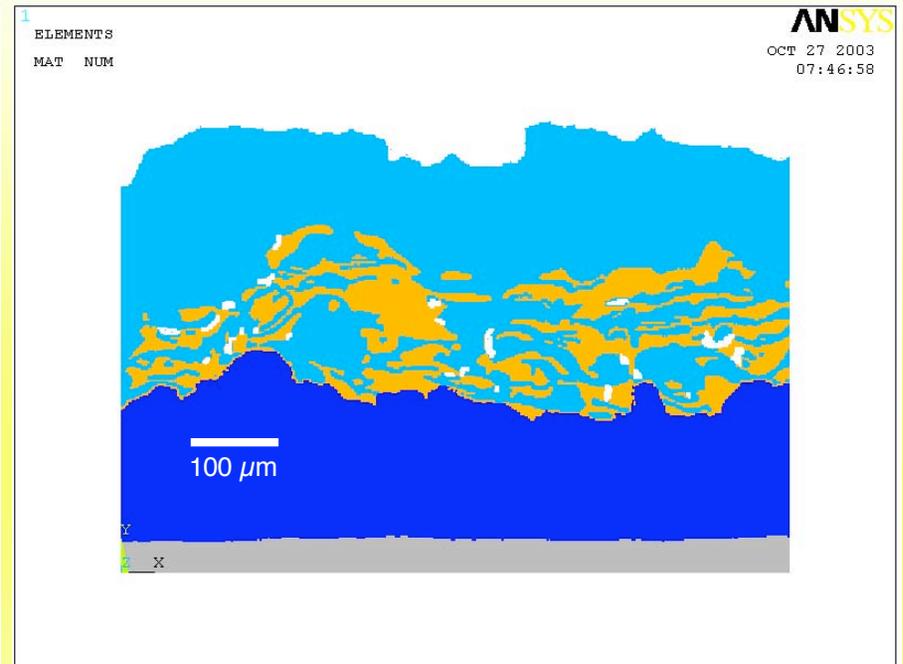


Case 1: σ_x Residual Stress Field Only

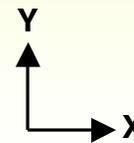
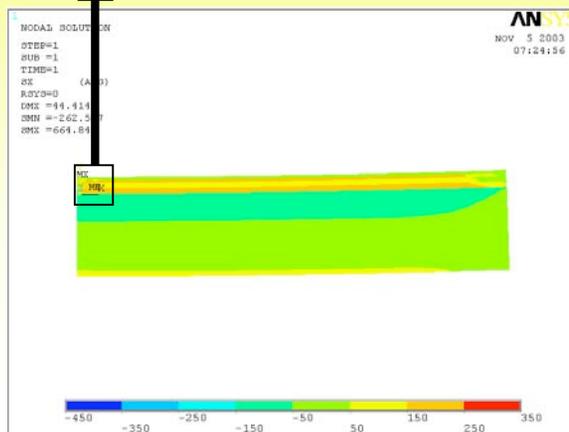
σ_x Stresses



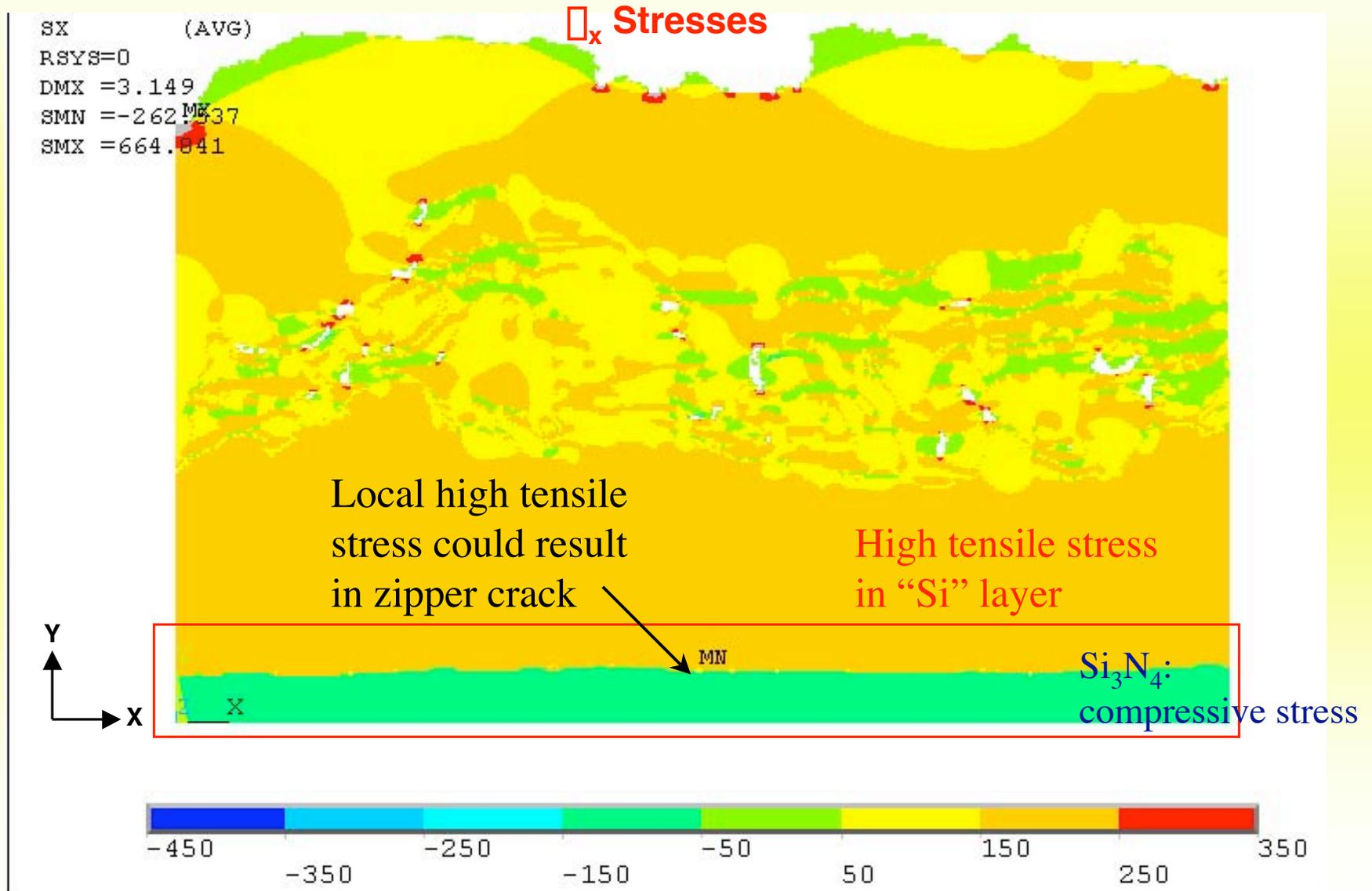
μ -FEA Reconstructed Model



σ_x Stresses in Whole Model

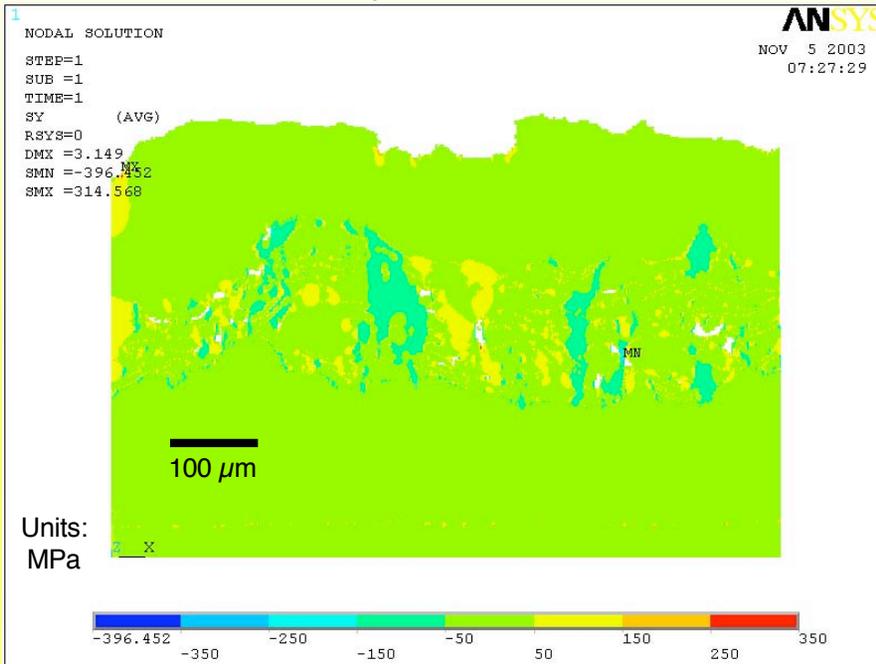


High Stress Concentration Observed at Si/Si₃N₄ Interface

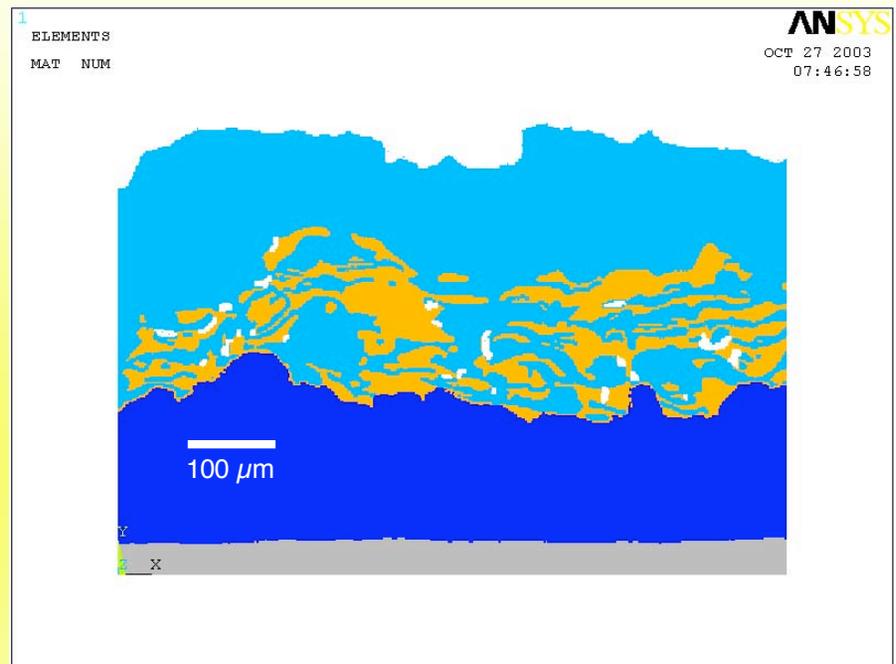


Case 1: σ_y Residual Stress Field Only

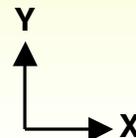
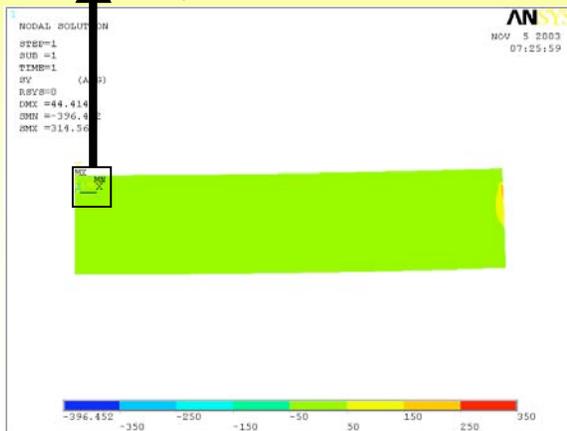
σ_y Stresses



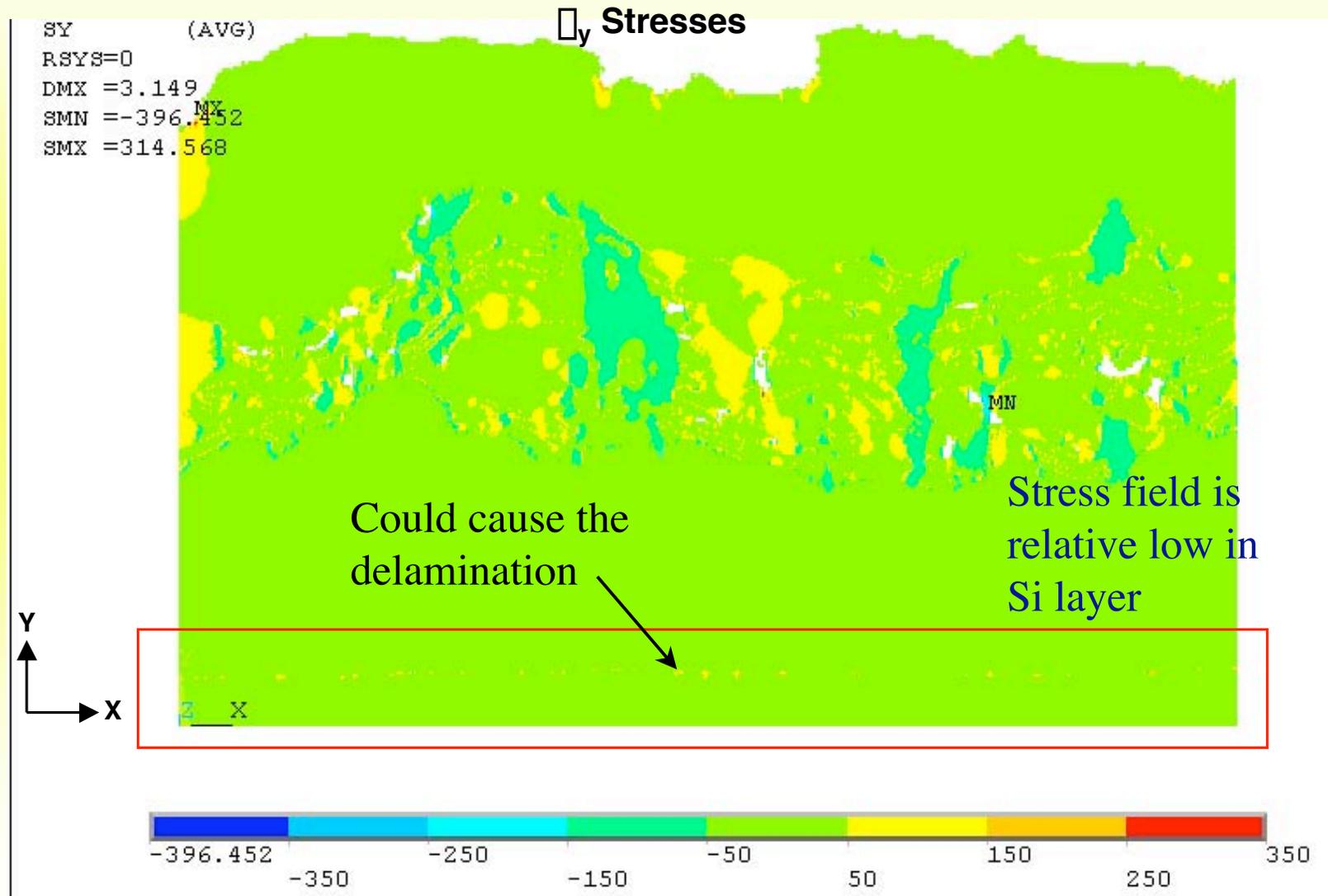
μ -FEA Reconstructed Model



σ_y Stresses in Whole Model

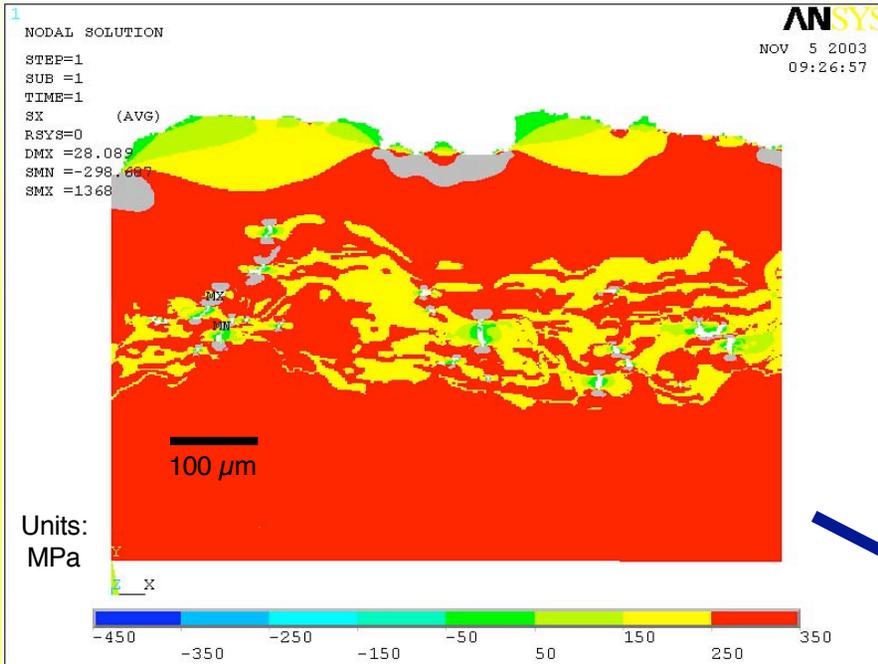


High Stress Concentration Observed at Si/Si₃N₄ Interface

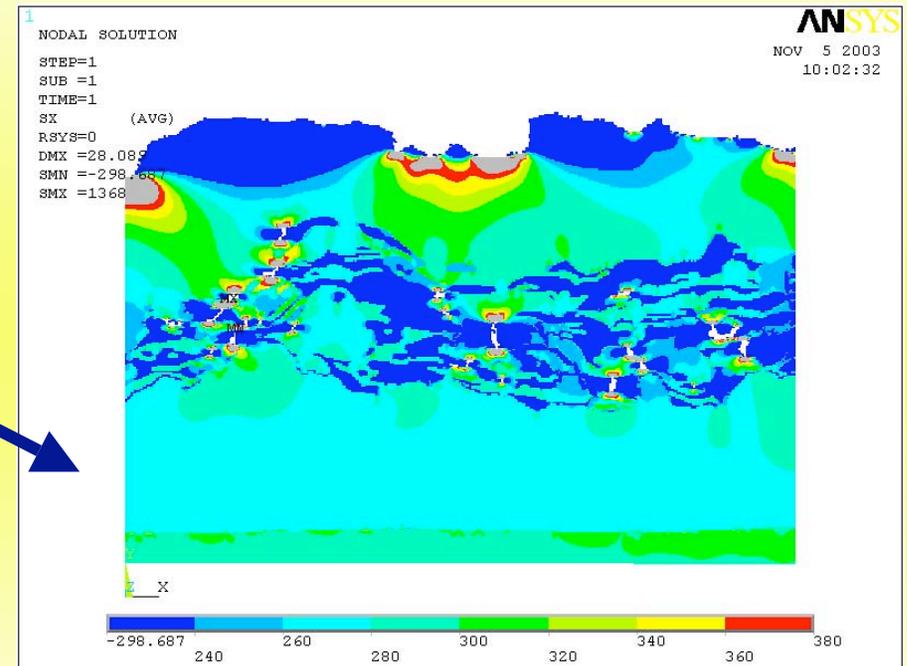


Case 2: Sum of σ_x Residual Stress Field + σ_x Bend Stress

σ_x Stresses

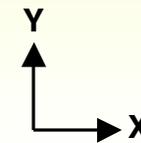
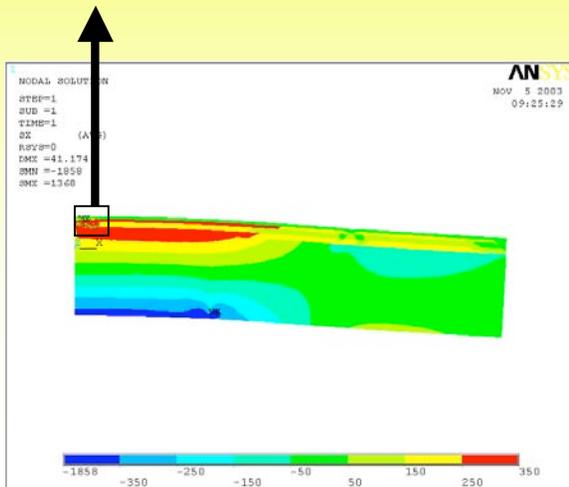


Refined View of σ_x Stresses

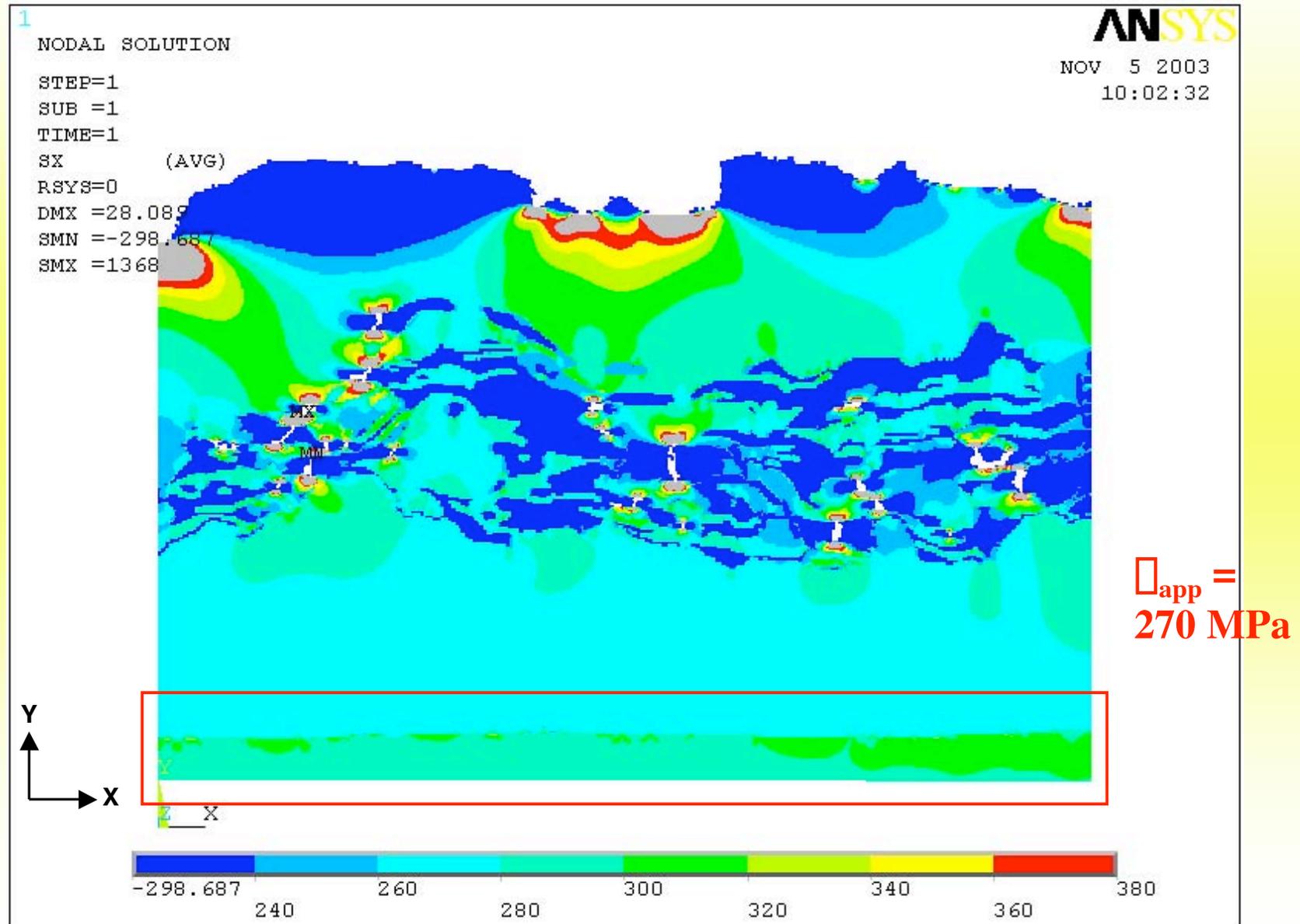


$\sigma_{\text{app}} = 270 \text{ MPa Bend Stress}$

σ_x Stresses in Whole Model

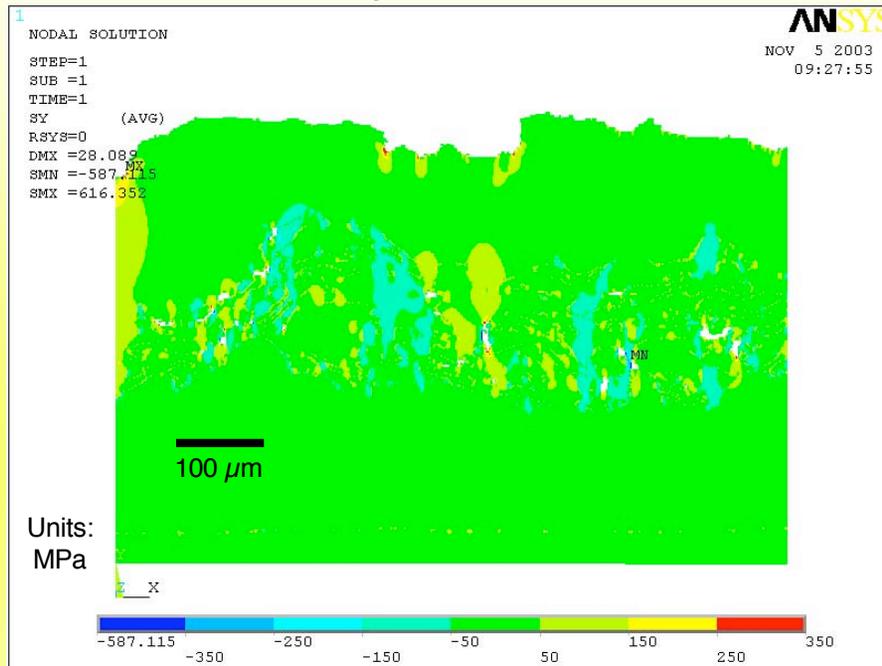


Externally Applied Stress Augments the Residual Stress Field

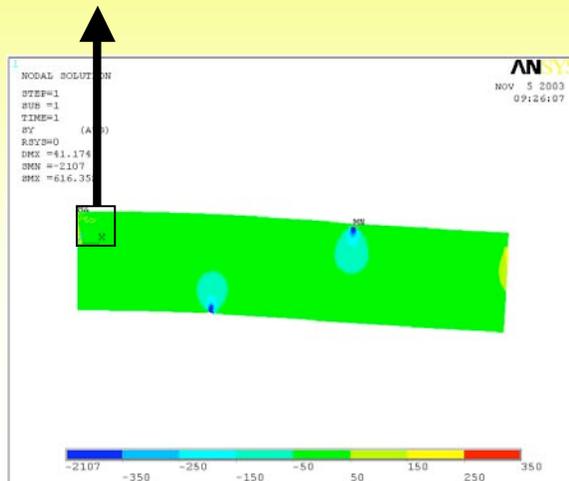
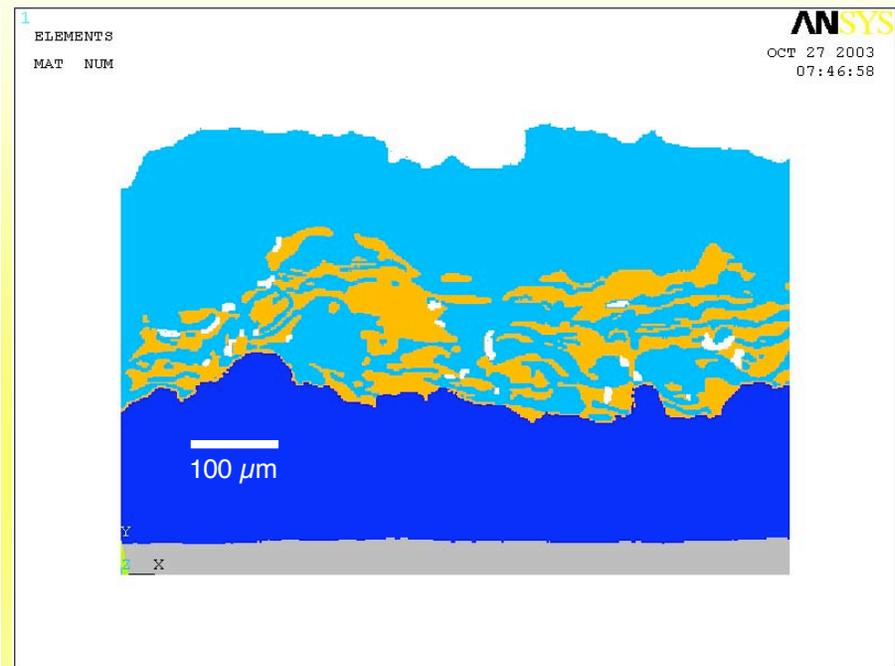


Case 2: Sum of σ_y Residual Stress Field + σ_y Bend Stress

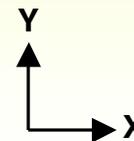
σ_y Stresses



μ -FEA Reconstructed Model



σ_y Stresses in Whole Model



Summary

- Grit blast conditions employed to prepare the component surface could impact the mechanical performance of materials (especially for low toughness ones).
- The biaxial disks and flexure MOR bars with EBC exhibited 67-75% low strength than those obtained from non-coated samples. Subsequent test and analysis suggests that strength degradation is less dependent on the thickness as well as surface roughness of Si bond coat.
- Observations of zipper cracks evident in Si_3N_4 substrates, suggestive failure initiation (and strength degradation) could be possibly associated with CTE mismatch between EBC and Si_3N_4 substrates.
- □-FEA results showed high residual stress field present in Si bond coat (~150-250 MPa) and Si/ Si_3N_4 interface (~50-150 MPa).
- Externally applied stress would augment the residual stress field and cause the premature fracture of ECB samples.